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A MECHANISTIC ANALYSIS OF HYDROGEN ENTRY INTO METALS

DURING CATHODIC HYDROGEN CHARGING

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A MECHANISTIC ANALYSIS OF HYDROGEN ENTRY INTO METALS DURING CATHODIC HYDROGEN CHARGING

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Introduction

Entry of hydrogen into metals is of serious concern to metallurgists and engineers, since it severely degrades the mechanical properties of metals (1). These problems arise in the cathodic protection of metals, power plants, and in environments where H_2S is present as in petroleum refining (2) where the hydrogen evolution reaction (h.e.r.) and hydrogen permeation reaction take place in the corroding or cathodically polarized metal. By performing hydrogen charging experiments of thin samples using the Devanathan-Stachurski cell, the permeation characteristics have been extensively studied (3,4). The present paper seeks to analyze the h.e.r. mechanism and to predict the relationship between the permeation flux and the charging and evolution (recombination) fluxes. A thorough development of the model and actual computations of rate constants and hydrogen coverages will appear elsewhere (5).

Analysis

Essentially, three steps are involved during cathodic hydrogen charging of metals. They are: (1) the hydrogen discharge reaction (proton tunneling), (2) hydrogen recombination reaction either by chemical recombination or electrochemical desorption, and (3) hydrogen permeation (mainly by bulk diffusion). Of these three, the bulk diffusion step is usually the slowest. A detailed schematic of the reactions is given in Fig. 1 below.

The charging current (i_c) is given by

$$i_c = i_0' (1 - \theta_s) e^{-\alpha \eta} \quad (1)$$

The hydrogen evolution current (i_r) (assuming chemical recombination of H atoms) is given by

$$i_r = F k_3 \theta_s^2 \quad (2)$$

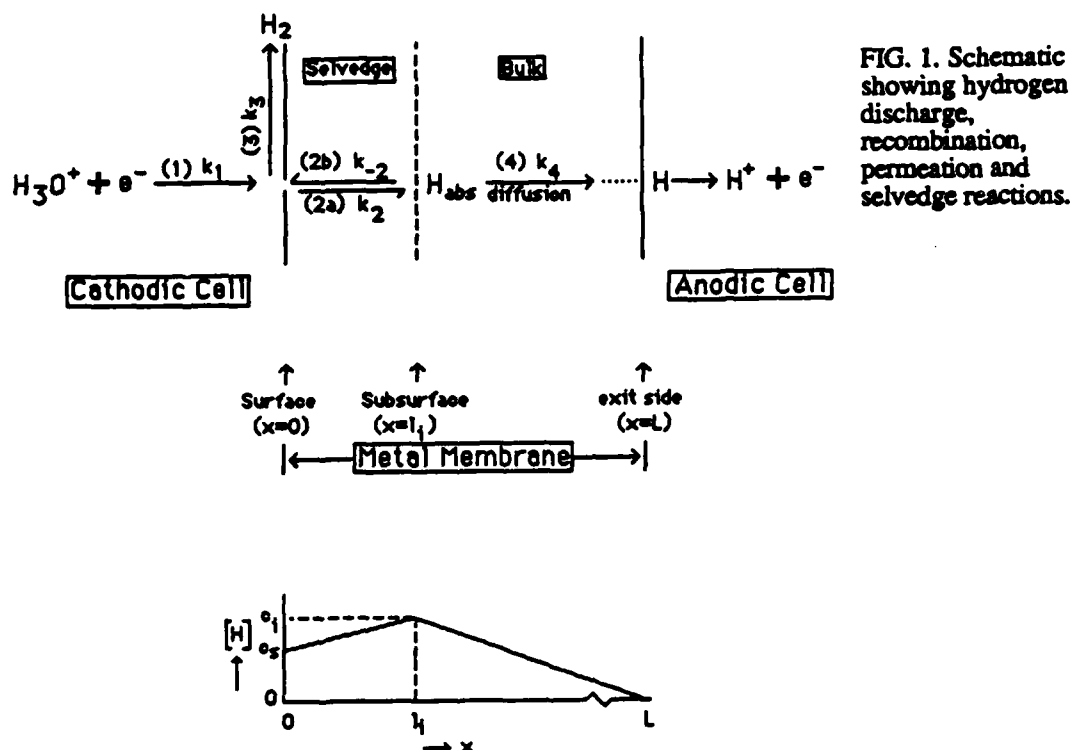


FIG. 1. Schematic showing hydrogen discharge, recombination, permeation and selvedge reactions.

The steady state hydrogen permeation current (i_{∞}) is given by

$$i_{\infty} \cong F \frac{D_1}{L} c_i \quad (3)$$

In equations 1, 2 and 3: $i_0' = Fk_1 = i_0/(1-\theta_e)$; i_0 = the exchange current density; θ_e = the equilibrium surface coverage of hydrogen; k_1 = the discharge rate coefficient = $k_1^0 c_{H^+} e^{-a\alpha E^{eq}}$ (acid) =

$k_1^0 e^{-a\alpha E^{eq}}$ (alkaline); k_1^0 = the rate constant for the forward reaction; $c_{H^+} = H^+$ ion concentration; $a =$

$F/RT = 38.94 \text{ (volts)}^{-1}$ at $T = 300K$; α = the transfer coefficient; E^{eq} = the equilibrium potential for the h.e.r; θ_s = the surface coverage of hydrogen; η = the hydrogen overvoltage = $E \text{ applied} - E^{eq}$; k_3 = the recombination rate coefficient; D_1 = the hydrogen diffusion coefficient in the metal; L = the membrane thickness.

The model considers a selvedge reaction (as a result of proton tunneling) that is quite fast and constitutes a transition layer of a thickness that could range upward from $\sim 1 \text{ nm}$ (that has yet to be determined by special experiments). This establishes a metal subsurface hydrogen concentration, c_i , at the boundary of the selvedge. Thus, hydrogen diffuses out to either surface, though mostly to the charging side to recombine to form H_2 molecules. An equilibrium will be established between the surface covered (adsorbed) hydrogen atoms and hydrogen just below the surface (in the adsorbed state, with concentration c_s). This equilibrium has been analyzed before (6,7) giving $\theta_s = c_s/k'$, where $k' =$

the equilibrium absorption - adsorption constant and $c_s = c_i - c_g$. (However, it is to be emphasized here that the selvedge reaction is not critical to the development and application of this model. Consideration of selvedge, on the other hand, helps to generalize the model; in the absence of a selvedge, $c_g = 0$ and $c_s = c_i$). Using these relations along with equations (1), (2) and (3) one can arrive at the following set of equations, which for the first time have taken into account the effect of i_{∞} on the h.e.r. kinetics:

$$i_{\infty} = \left[\frac{k'}{b\sqrt{Fk_3}} \right] \sqrt{i_r} + \frac{c_g}{b} \quad (4)$$

and

$$i_c e^{\alpha \eta} = - \left(\frac{bi_o'}{k'} \right) \left(i_{\infty} - \frac{c_g}{b} \right) + i_o' \quad (5)$$

where $b = L/(FD_1) =$ a constant for a metal; also,

$$\left| -120 \frac{\text{mV}}{\text{decade}} \right| \sim \left| \frac{d\eta}{d \log i_c} \right| < \left| \frac{d\eta}{d \log i_{\infty}} \right| < 2 \left| \frac{d\eta}{d \log i_c} \right| < \left| -240 \frac{\text{mV}}{\text{decade}} \right| \quad (6)$$

for the model.

The transfer coefficient is given by

$$\alpha \equiv - \left(\frac{d \ln i_c}{d \eta} \right) / a \quad (6a)$$

Details of the derivations are shown elsewhere (5).

The potential range (η_c^l to η_c^u) in which the recombination reaction will be coupled with the discharge reaction (commonly observed on many metals) is given by (5)

$$\eta_c^l = [\ln (10k_1/k_3)] / (\alpha \alpha) \quad (7a)$$

$$\eta_c^u = [\ln (k_1/(10k_3))] / (\alpha \alpha) \quad (7b)$$

where the superscripts l and u refer to the lower and upper limits of the overpotential range.

Results and Discussion

In hydrogen permeation experiments, i_c is set and when the permeation current becomes independent of time, i_{∞} is measured. Then $i_r = i_c - i_{\infty}$.

If the plots of i_{∞} vs $\sqrt{i_r}$ (equation (4)) and $i_c e^{\alpha \eta}$ vs $[i_{\infty} - \frac{c_g}{b}]$ (equation (5)) are linear, then all of the coefficients k' , k_3 , c_g and i_o' can be calculated. Such calculations have been done on experimental data from the literature for iron and nickel membranes and found to verify the model (5) since these two plots are linear. An example of such an analysis is given for the polarization and



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permeation data of Bockris et al (3) obtained on Armco iron in 0.1N H_2SO_4 solution. Fig. 2 shows the plot of i_{∞} vs $\sqrt{i_r}$ and Fig. 3 shows $i_c \exp(\alpha F \eta) / RT$ vs $[i_{\infty} - \frac{C_H}{b}]$ plot for the data of Bockris et al (3). It is easily seen that these plots are linear and hence the model can be applied to determine the rate constants and exchange current density (Table 1). Then, from Eqn. (2), θ_s can be calculated using the k' and i_0 values obtained from the slope and intercept, respectively, of Fig. 3, the k_3 value obtained from the slope of Fig. 2 and the i_r value from $i_c - i_{\infty}$. The surface coverage (θ_s) vs the hydrogen overvoltage (η) plot (for the Bockris et al data) is shown in Fig. 4. It can be seen that the coverage is quite low in the potential range of experimentation.

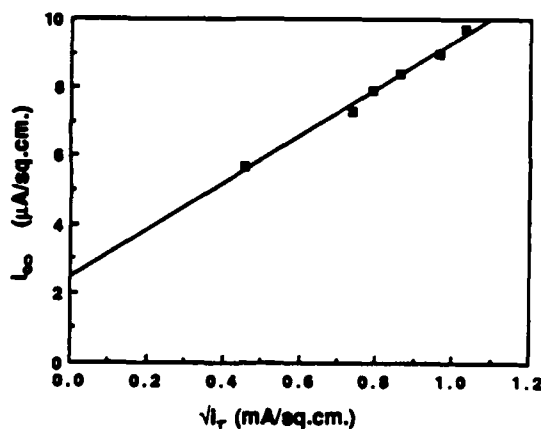


FIG. 2. Analysis of Data of Bockris et al (3).

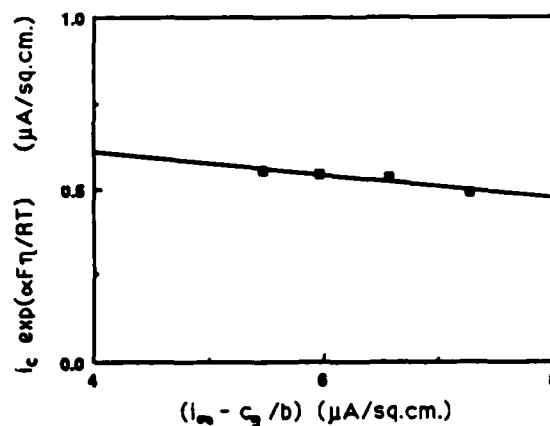


FIG. 3. Analysis of Data of Bockris et al (3).

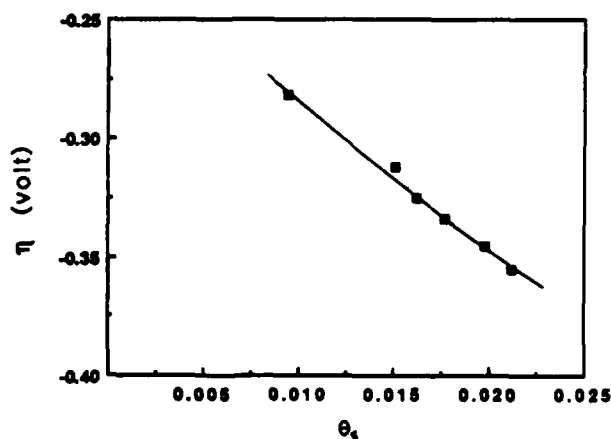


FIG. 4. H Coverage for the Data of Bockris et al (3).

TABLE 1
Calculated Values by Applying the
Model to the Data of Bockris et al (3)

$i_0 = 0.5 \mu A/sq.cm.$
$k_1 = 5.5 \times 10^{-12} \text{ mol} / \text{cm}^2.s$
$k_3 = 2.3 \times 10^{-5} \text{ mol} / (\text{cm}^2.s$
$k' = 2.6 \times 10^{-5} \text{ mol} / \text{cm}^3$
$\eta_c^l = -600 \text{ mV} ; \eta_c^u = -810 \text{ mV}$

Thus, this model effectively accounts for the contribution of i_{∞} to the overall kinetics. The most important prediction of this model is that i_{∞} is proportional to $\sqrt{i_r}$ and not to $\sqrt{i_c}$ as assumed in earlier models (3). This type of relationship ($i_{\infty} \propto \sqrt{i_r}$) has been previously observed (8).

The above relationships assume that $\eta \gg RT/F$ so that the backward reactions can be neglected. Also, the Langmuir isotherm of hydrogen coverage was utilized in order to simplify the derivation. However, in many cases, the reactions (discharge and recombination) are activated in which case Frumkin-Temkin corrections (9) have to be applied for θ_s in the equations for i_c and i_r . Equation (1) then becomes:

$$i_c = i_0 (1 - \theta_s) e^{-\alpha f \theta_s} e^{-\alpha \eta} \quad (8)$$

and equation (2) becomes

$$i_r = F k_3 \theta_s^2 e^{2\alpha f \theta_s} \quad (9)$$

where $f = \gamma/RT$, γ being the gradient of the apparent standard free energy of adsorption with coverage. The value of $f = 4$ to 5 for H coverages (9). In the problem of enhanced hydrogen entry in the presence of H_2S , such considerations have been shown to be necessary (10). The modified relationships between i_{∞} , i_r and i_c are given by

$$\ln \left(\frac{\sqrt{i_r}}{i_{\infty}} \right) = \left(\frac{\alpha f b}{k'} \right) i_{\infty} - \ln \left(\frac{k'}{b \sqrt{F k_3}} \right) \quad (10)$$

and

$$\ln (f(i_c, i_{\infty})) = \alpha a (-\eta) + \ln (i_0') \quad (11)$$

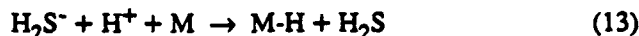
where

$$f(i_c, i_{\infty}) = \frac{i_c e^{\frac{(\alpha f b i_{\infty})}{k'}}}{\left(1 - \frac{b i_{\infty}}{k'}\right)}$$

Thus, equation (10) tells us that i_{∞} will not be linearly related to $\sqrt{i_r}$ when $f > 0$, meaning the discharge and recombination reactions are activated, probably due to a side reaction of H_2S with a hydrated electron, e^-_{aq} (10), as follows:



and



If $\ln \left(\frac{\sqrt{i_r}}{i_{\infty}} \right)$ vs i_{∞} and $\ln (f(i_c, i_{\infty}))$ vs η are linear, then this side reaction and overall mechanism can be

said to be operating. Then, the coefficients k' , k_3 , α and i_0' can be computed from the slopes and intercepts of these plots, in conjunction with the iterative solution of equations (10) and (11).

Eventually, θ_s vs η can be plotted; and θ_e , i_0 and k_1 can be computed.

Once again, the potential ranges, where the recombination and discharge reactions are coupled, can be estimated from equations (7a) and (7b). With increasing H_2S concentration, these potential ranges have been found from analysis of available data according to the above model to become less negative, k_3 progressively decreases suggesting a decreasing surface diffusivity of H_{ad} atoms, and k_1

progressively increases suggesting that the discharge reaction is enhanced by the side reaction, equation (12), followed by equation (13). This will also explain why the overvoltage actually decreases rather than increases in the presence of H_2S .

Conclusions

A summary of a recently completed analysis of hydrogen electrode reactions during aqueous cathodic charging of metals is presented. The analysis for the first time takes into account the effect of hydrogen permeation into the metal on the h.e.r. From the model all of the kinetic parameters are computable without use of any adjustable parameters. For the first time, surface coverages and rate constants are determinable from the measured charging and permeation currents. The enhancement of hydrogen entry in the presence of poisons, such as H_2S , can be analyzed with the model, taking into account the Frumkin-Temkin isotherms in the discharge and recombination reaction kinetics.

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References

1. M. Smialowski, Hydrogen in Steel, Pergamon Press, Oxford (1962).
2. C. M. Hudgins, Materials Protection, Vol. 8, p. 41 (1969).
3. J. O'M. Bockris, J. McBreen and L. Nanis, J. Electrochem. Soc., Vol. 112, p. 1025 (1965).
4. M. Zamanzadeh, A. Allam, H. W. Pickering and G. K. Hubler, J. Electrochem. Soc., Vol. 127, No. 8, p. 1688 (1980).
5. R. N. Iyer, M. Zamanzadeh and H. W. Pickering "Analysis of Hydrogen Evolution and Entry into Metals for the Coupled Discharge-Recombination Mechanism," (submitted to the J. Electrochemical Society).
6. C. D. Kim and B. E. Wilde, J. Electrochem. Soc., Vol. 118, p. 202 (1971).
7. B. E. Wilde and C. D. Kim, Corrosion, Vol. 42, No. 4, p. 243 (1986).
8. E. G. Dafft, K. Bohnenkamp and H. J. Engell, Corrosion Science, Vol. 19, p. 591 (1979).
9. E. Gilcadi and B. E. Conway, Modern Aspects of Electrochemistry, No. 3, J. O'M. Bockris and B. E. Conway eds., Butterworths, Washington (1964), pp. 347-442.
10. R. N. Iyer, I. Takeuchi, M. Zamanzadeh and H. W. Pickering, "Hydrogen Sulfide Effect on Hydrogen Entry in Iron - A Mechanistic Study" (to be submitted to Corrosion).

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